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Article

Concept of Triphenylamine Main Chains with Dual Electroactive Nitrogen Centers toward Record-High Stable Electrochromic **Polyamides**

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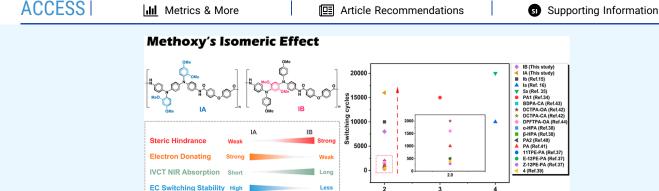


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ABSTRACT: Two novel electrochromic (EC) polyamides (PAs) with dual electroactive nitrogen sites, IA and IB, were synthesized from 4,4'-dicarboxydiphenyl ether with N,N'-bis(4-aminophenyl)-N,N'-bis(2,4-dimethoxyphenyl)benzene-1,4-diamine (2) and N,N'-bis(4-aminophenyl)-N,N'-bis(4-methoxyphenyl)-2,5-dimethoxybenzene-1,4-diamine (4), respectively. IA demonstrated superior EC performances, including multiple color changes, intense absorption in the near-infrared (NIR) region, fast switching speed, and exceptional EC stability (after 16,000 switching cycles in the first oxidation stage, only 4.0% coloration efficiency (CE) decay at 1016 nm). The high EC stability of IA could originate from the resonance effect between the different redox states and the effect of electron-donating dimethoxy substituents. Although the isomeric IA and IB contain the same number of electron-donating methoxy substituents and electroactive nitrogen centers, IA+ is significantly more stable than IB+ due to the steric hindrance of the ortho-substituents (methoxy groups) at the electroactive nitrogen centers, which hinders the resonance effect of the electroactive centers for IB⁺. The steric hindrance for IB increases the oxidation potential and a broader NIR absorption. Notably, the resonance effect of the oxidized electroactive centers plays a crucial role in stabilizing cation radicals. Moreover, IA and IB exhibit strong absorption properties in the NIR and visible regions in the first oxidation state, indicating their substantial potential applications in smart windows, EC displays, and other high-performance optoelectronic devices.

KEYWORDS: triphenylamine, electrochromism, high stability, substitution effect, mixed valence

INTRODUCTION

Electrochromic (EC) materials are defined by their ability to undergo a reversible optical change in absorption upon electrochemically oxidized or reduced. 1,2 These materials are typically categorized into inorganic and organic chromophores. The first observation in inorganic EC materials, such as tungsten trioxide (WO₃) and Prussian blue, is widely utilized due to their excellent durability and good thermal stability.³ However, they are limited in their ability to adjust the colors. The organic EC materials (e.g., viologens and conducting polymers) have gained considerable attention for switching among their different redox states to tune the colors. Conducting polymers offer several advantages over inorganic compounds, including outstanding CE, high optical contrast,

fast switching ability, good processability, and multiple color variations.4-10

Among organic EC materials, π -conjugated polymers have emerged as highly versatile structures due to their adjustable donor-acceptor structures and extended electronic delocalization. For instance, carbazole-based conjugated polymers bearing substituents exhibit rapid switching (e.g., <1 s), high

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Scheme 1. Synthesis of Compounds 1 to 4

CE (>150 cm²/C), and significant optical contrast (>40%) in the near-infrared (NIR) region. 11 Kim et al. also developed donor—acceptor structured conjugated polymers exhibiting exceptional coloration efficiencies (>200 cm²/C), fast response times (<1 s), and excellent cycling stability, making them suitable for multifunctional smart window applications. 12 These outstanding EC performances are attributed to the effective resonance stabilization and planar backbones of the molecular structures, which facilitate efficient charge delocalization and suppress structural distortion during redox processes. These findings underscore the critical roles of resonance and steric effects in achieving long-term stability and high efficiency in electrochromic behavior.

Triphenylamine (TPA)-based polymers have attracted significant attention due to their optical transparency and colorlessness in the neutral state and intriguing EC behaviors, including high CE, quick switching responsiveness, good filmforming properties, and multiple color variations. 13-18 In recent years, when the triphenylamine (TPA) structure has only one electroactive center, its stability is usually lower. The main reason is that the free radical cations formed after oxidation lack an effective delocalization and stabilization mechanism, resulting in the inability to effectively disperse the charge, and thus, the degradation of electrochemical stability in long-term or repeated redox cycles.¹³ However, dual-band electrochemical materials can effectively solve the problem of poor stability, and they have good control ability in the visible light and near-infrared spectroscopy, as well as satisfactory energy-saving efficiency, so they have attracted extensive

attention.¹⁹ These materials hold elevated potential for applications in smart windows, nonemissive displays, and adaptive camouflage.^{15,20–22} For example, EC windows could mitigate the transmitted NIR radiation, potentially lowering energy consumption.²³ The mixed-valence (MV) compounds were classified by Robin and Day into three types based on the different electron coupling strengths.²⁴ TPA-based polymers with dual electroactive nitrogen centers have been reported as a symmetrical delocalized class II or III intervalence charge transfer (IV-CT), leading to a characteristic NIR absorption band.²⁵ Thus, this unique absorption could facilitate dual-band EC materials to achieve adequate energy-saving benefits.

EC materials are required to fulfill several critical criteria for practical applications, including rapid switching, substantial optical transmittance variation, high coloration efficiency (CE), and long-term stability. Among these, long-term electrochemical stability is widely recognized as one of the most challenging and essential properties for real-world implementation.²⁶ TPA-based polymers have attracted significant attention as mentioned previously; however, these systems often suffer from performance degradation under prolonged redox cycling. Notably, most reported TPAcontaining polymers exhibit approximately a 10% CE decrement after only 2000 EC switching cycles. 14,16,27-31 To mitigate this limitation, recent studies have explored the incorporation of multiple redox-active nitrogen centers within the polymer backbone to promote charge delocalization and enhance the stability of the oxidized radical cations.³²⁻³⁵ For instance, Yen and Liou developed a series of solution-

processable TPA-based aromatic polyamides, in which flexible ether linkages and efficient intramolecular charge delocalization afforded high CE values (388 cm²/C), strong nearinfrared (NIR) electrochromism ($\lambda_{max} = 1080$ nm), and excellent stability over 10,000 redox cycles. Similarly, starshaped triarylamine-containing polymers designed by Liou et al. demonstrated rapid switching times (<2 s), high CE (290 cm²/C), and minimal electrochromic degradation, underscoring the favorable impact of multibranched conjugated architectures on performance and durability. 16 Nevertheless, increasing the number of electroactive centers within TPAcontaining structures typically involves complex synthetic pathways and multistep monomer design, thereby limiting the scalability and practical feasibility of such approaches. As a result, achieving high EC stability using structurally simpler systems with fewer electroactive centers remains a significant challenge. Recent findings have highlighted the importance of redox center positioning within the polymer structure. Specifically, TPA-based polymers bearing para-methoxyphenyl substituents with dual electroactive centers in the main chains exhibit markedly superior EC stability than their isomeric counterparts with similar substituents positioned in the side chains. For example, a main-chain-configured TPA polymer maintained >90% of its initial CE after 10,000 cycles, whereas a structurally related side-chain analogue showed a CE loss of 5.6% after only 1000 cycles. 15,36 These observations suggest that main-chain-oriented conjugation facilitates more effective electron delocalization and redox stabilization.

Recently, several researchers designed and synthesized TPAbased polymers with dual electroactive centers, but they still show only moderate electrochemical or EC stability.3 Therefore, developing dual-electroactive centered TPA-based polymers on the main chain is worthwhile for further elevating the EC stability. Hence, this study focuses on exploring organosoluble TPA-based polyamides (PAs) with exceptional EC stability, high CE, fast switching capability, and two-stage color changes. Among the characteristics of EC performance, obtaining gratifyingly high long-term EC stability of the TPAbased polymers is relatively more challenging. Typically, cation radicals would undergo resonance on the para and ortho positions to the electroactive nitrogen center. As aforementioned, bearing a para-methoxyphenyl substituent can efficiently enhance the EC stability. 13,15 As a result, this study highlights that structural design, particularly incorporating more electron-donating methoxy groups in the ortho position, is crucial for improving the electrochemical stability of polymers. Herein, two novel EC polyamides with dual electroactive nitrogen sites, IA and IB, were prepared from 4,4'-dicarboxydiphenyl ether and two newly synthesized diamine monomers, IA and IB, containing the same number of electron-donating methoxy groups in the repeating unit, while attaching the additional two methoxy groups on different ortho positions of the electroactive diamine's pendant phenyl group or the phenyl π bridge. Furthermore, this study also elucidated the effects of the substituent position on EC performance by designing the redox-active TPA-based isomeric polyamide structures.

RESULTS AND DISCUSSION

Monomer Synthesis. Diamine monomer **2** was synthesized via two reaction steps, which start from N,N'-bis(4-nitrophenyl)benzene-1,4-diamine, as illustrated in Scheme 1. The C-N coupling reaction was carried out by reacting N,N'-

bis(4-nitrophenyl)benzene-1,4-diamine with 1-iodo-2,4-dimethoxybenzene to produce intermediate 1, which was hydrogenated to yield the new diamine monomer 2. Similar procedures were used to prepare the new diamine monomer 4. The C-N coupling reaction was carried out by reacting 1,4dibromo-2,5-dimethoxybenzene with 4-methoxy-4'-nitrodiphenylamine in the presence of copper to produce intermediate 3 and then hydrogenated to prepare the new diamine 4. The lower yield of compound 3 may be attributed to the use of a less reactive bromide (1,4-dibromo-2,5-dimethoxybenzene) in the C-N coupling reaction, instead of a more reactive iodide, and the melting point of compound 3 is higher than that of the other three compounds. We speculate that it may be related to a different configuration during crystallization than the other compounds. In addition, the relatively low yields of 2 and 4 are presumably due to the use of DMF for recrystallization. FT-IR and NMR spectroscopies were employed to characterize the chemical structures of synthesized compounds 1-4. The FTIR spectra depicted in Figures S1 and S2 shows that the characteristic nitro group absorption bands of 1 and 3 were observed at approximately 1597/1591 cm⁻¹ (asymmetric stretching) and 1309/1309 cm⁻¹ (symmetric stretching), respectively. These nitro bands disappeared after reduction to compounds 2 and 4, and primary amino group bands appeared at 3427/3427 cm⁻¹ and 3350/3346 cm⁻¹ (N-H stretching), respectively. ¹H and ¹³C NMR spectra, as shown in Figures S5-S12, are in good agreement with the proposed molecular structures of compounds 1-4. 4.72 and 4.86 ppm signals are characteristic of monomers 2 and 4 amino groups, respectively (Figures S7 and S11). The NMR and IR spectra confirmed all of the compounds reported herein. In addition, the structure of dinitro compound 3 was confirmed through single-crystal X-ray diffraction, and the single crystal was obtained via slow crystallization from DMF. As shown in Figure 1, compound 3 exhibits a propeller-shaped configuration around the TPA core, showing noncoplanar orientations, indicating that the entire structure adopts a twisted conformation.

Polymer Synthesis. According to the phosphorylation method described by Yamazaki, 45,46 two new PAs, **IA** and **IB**, were synthesized from the diamine monomer 2 and 4 with

Figure 1. X-ray structure of 3

4,4'-dicarboxydiphenyl ether. The chemical structures of polyamides IA and IB are presented in Scheme 2. All

Scheme 2. Chemical Structures of Polyamides IA and IB

Referenced Polyamide

polymerization reactions proceeded homogeneously, resulting in high inherent viscosities of 0.8 and 0.5 dL/g for PAs IA and IB, respectively, as summarized in Table S1. The observed difference in inherent viscosities might be attributed to the variation in molecular rigidity and steric hindrance between the two polyamides. Specifically, the ortho-substituted structure of IB induces greater dihedral twisting and steric congestion, which can hinder chain growth and reduce the molecular weight, leading to a lower viscosity. In contrast, the more planar structure of IA facilitates better chain packing and entanglement, resulting in a higher inherent viscosity. All of the polymers were successfully prepared as transparent and tough films (Figure S15) through drop casting. IR and ¹H NMR spectroscopies confirmed the resulting polyamides. The IR spectra (Figures S3 and S4) exhibited characteristic amide absorption peaks at around 3316 and 3303 cm⁻¹ (amide N-H stretching), as well as 1654 and 1690 cm⁻¹ (amide carbonyl) for IA and IB, respectively. Furthermore, the ¹H NMR spectra verified the chemical structures of IA and IB, with the proton assignments detailed in Figures S13 and S14.

SOLUBILITY AND THERMAL PROPERTIES

Table S1 presents the prepared PAs' solubility in various organic solvents. IA and IB showed good solubility in the tested solvents, including DMAc, DMF, NMP, m-cresol, DMSO, and o-chlorophenol. However, IA shows a higher inherent viscosity (0.80 dL/g) and weight-average molecular weight $(M_w = 71.5 \text{ kDa}; \text{ PDI} = 2.03)$ compared to IB (0.50) dL/g; 49.7 kDa; PDI = 1.87), indicating a greater degree of chain entanglement and molecular weight development. These results suggest that the more planar structure of IA facilitates intermolecular interactions and promotes chain growth during polymerization. Besides, when observing the chemical shifts of the amino group in ¹H NMR spectra, it is found that diamine 4 (4.72 ppm) exhibited a more upfield shift than diamine 2 (4.86 ppm), implying that a more basic diamine unit to undergo polymerization. Thus, the resulting polyamide IA would demonstrate a higher molecular weight and inherent viscosity than IB. The excellent solubility of these PAs makes them highly suitable for practical applications, such as inkjet printing and spin-coating, facilitating the fabrication of thin films for optoelectronic devices.

The thermal properties of the PAs were analyzed using thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC), with the results summarized in Table S2. The TGA curves of both IA and IB revealed remarkable thermal stability with no obvious weight loss before 400 °C, as shown in Figure S16. The temperatures at 10% weight loss in air and nitrogen of IA and IB ranged between 405 and 434 and 403-418 °C, respectively, indicating good thermal stability. The residual wt % (char yield) of IA and IB in a nitrogen atmosphere was more than 51% at 800 °C. The glass-transition temperatures (T_{σ}) of PAs IA and IB could be easily measured in the DSC thermograms, as shown in Figure S17. The resulting polymers exhibit relatively high $T_{\rm g}$ values, which Ib, IA, and IB revealed as 236, 235, and 215 °C, respectively. According to the simulation results (Figure 2), the dihedral angles of the model units were found to follow this sequence: IA-MU $(40.47^{\circ}) \sim \text{Ib-MU} (41.05^{\circ}) < \text{IB-MU} (52.42^{\circ})$. IB-MU exhibits a larger torsion angle than IA-MU and Ib-MU, resulting in a more twisted backbone that hinders efficient polymer chain packing and leads to a lower glass transition temperature. In contrast, IA, with its smaller dihedral angle and more planar backbone, enables tighter molecular packing, thereby enhancing the thermal stability. On the other hand, the methoxy groups on the π -bridge of **IB** introduce steric hindrance that distorts the backbone conformation and disrupts $\pi - \pi$ stacking and intermolecular hydrogen bonding, compromising thermal stability. These thermal performance data further confirm the suitability of PAs IA and IB for hightemperature applications, particularly in optoelectronic devices that demand superior thermal stability and mechanical performance.

Electrochemical Properties. The cyclic voltammetry (CV) were assessed to evaluate the electrochemical properties of the polyamides deposited onto indium—tin oxide (ITO)-coated glass slides. The CV profiles for polyamides **IA** and **IB** are illustrated in Figure 3.

Sas shown in Figure 4a, the CV scans from -0.20 to 0.60 V revealed an exceptionally stable redox process over 20,000 cycles, with only a barely noticeable decrease in the peak current. Moreover, the characteristic absorption peak at 342 nm exhibits an excellent 98.2% reversibility over 20,000 cycles (Figure 4b). However, as

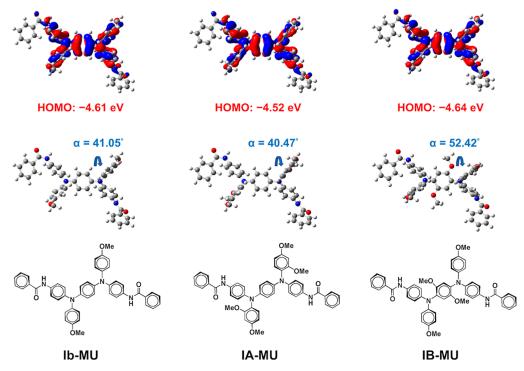


Figure 2. Calculated HOMO levels of the model units (MUs) of Ib, IA, and IB using DFT at the B3LYP/6-31G(d) level.

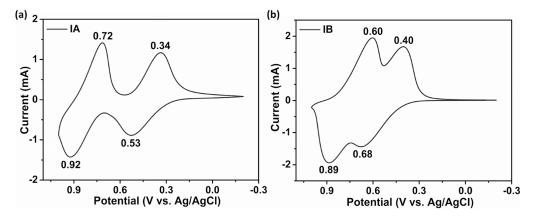
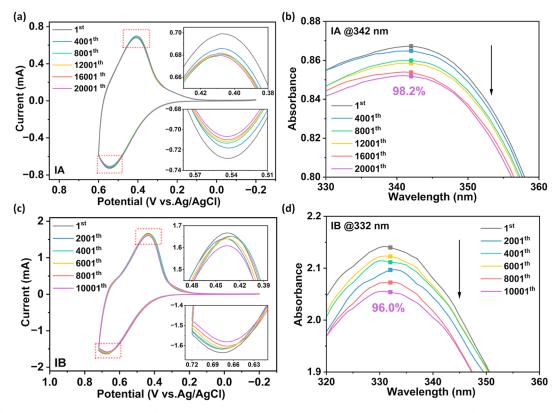


Figure 3. Cyclic voltammetric diagrams of PA films (a) IA (film thickness: 250 ± 30 nm) and (b) IB (film thickness: 250 ± 30 nm) measured on the ITO-coating (0.6 cm \times 2.5 cm) glass substrate in 0.1 M TBAP/MeCN at a scan rate of 50 mV/s.

depicted in Figure 4c,d, the reversibility of IB could only retain 96.0% after 10,000 cycles, whereas IA demonstrated superior stability than IB at the 10,000th cycle. Therefore, IB did not do further testing to 20,000 cycles. This difference mainly comes from the different π -bridges in their main chains. The π bridge in IA consists of simple benzene rings, which provide good conjugation and planarity and help the radical cation resonate easily along the main chain, making the oxidized form more stable. In contrast, the π bridge in IB contains methoxy groups that twist the conformation. As a result, IB revealed inferior stability to IA.⁴⁷

The observed redox potentials and the calculated highest occupied (HOMO) and lowest unoccupied molecular orbital (LUMO) levels of **IA** and **IB** are summarized in Table 1. The HOMO levels of **IA** and **IB**, estimated from the $E_{1/2}$ values, were -4.80 eV and -4.90 eV, respectively. Comparing homologues **IA** and **Ib** containing dual electroactive centers, we found that **IA** exhibited a lower oxidation potential than **Ib** (**IA**: $E_{\text{onset}} = 0.30$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V, $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{\text{onset}} = 0.42$ V; $E_{1/2} = 0.44$ V; **Ib**: $E_{1/2} = 0.44$ V; **Ib**:

0.54 V) because IA contains more effective electron-donating methoxy substituents in the repeating unit than does Ib. As a result, IA exhibits a higher electron density on the electroactive nitrogen of the TPA unit due to the electron-donating effect, resulting in lower E_{onset} and $E_{1/2}$. Comparing homologues IA and IB, IB exhibited higher oxidation potential than IA (IB: $E_{\text{onset}} = 0.42 \text{ V}, E_{1/2} = 0.54 \text{ V}; \text{ IA: } E_{\text{onset}} = 0.30 \text{ V}, E_{1/2} = 0.44$ V). Although IA and IB contain the same number of electrondonating methoxy substituents in the repeating unit, IB showed a higher oxidation potential, resulting from the relatively high steric hindrance of the ortho-substituents (methoxy groups) at the electroactive nitrogen center which affects the resonance effect between the electroactive centers. J. Reynolds et al. also reported that steric repulsion between substituents on adjacent rings increases oxidation potentials. 48,49 In addition, theoretical calculations were employed to clarify the results obtained from the electrochemical studies (Figure 2). Using the HOMO energy of **Ib-MU** (-4.61 eV) as the reference, the addition of methoxy substituents to pendant



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Figure 4. (a) CV stability diagram of IA film (thickness: 250 ± 30 nm) at the first oxidation stage (-0.20 to 0.60 V) for 20,000 cycles. (b) UV-vis spectra of IA film in the neutral state after different scanning cycles. (c) CV stability diagram of IB film (thickness: 250 ± 30 nm) at the first oxidation stage (-0.20 to 0.72 V) for 10,000 cycles. (d) UV-vis spectra of IB film in the neutral state after different scanning cycles.

Table 1. Electrochemical and Optical Properties of Resulting PAs

	oxidation potential (V)							
	film (nm)			$E_{1/2}^{c}$				
code	λ_{\max}^{a}	$\lambda_{ m onset}^{a}$	$E_{\rm g} ({\rm eV})^b$	$E_{ m onset}$	1st	2nd	HOMO (eV) ^d	LUMO (eV) ^e
IA	342	405	3.06	0.30	0.44	0.82	-4.80	-1.84
IB	332	402	3.08	0.42	0.54	0.75	-4.90	-1.82
\mathbf{Ib}^f	342	411	3.02	0.42	0.54	0.90	-4.86	-1.84

"UV/vis absorption measurements are in the film state. Calculated in the film state by the equation: $E_g = 1240/\lambda_{onsev}$ where E_g is the energy gap between HOMO and LUMO. Versus Ag/AgCl in MeCN in the cyclic voltammetry. $E_{1/2}$: Average potential of the redox couple peaks. Calculated from cyclic voltammetry using ferrocene/ferrocenium as reference (-4.80 eV and $E_{1/2} = 0.44$ V). LUMO = HOMO + E_g . Data were cited from ref 15.

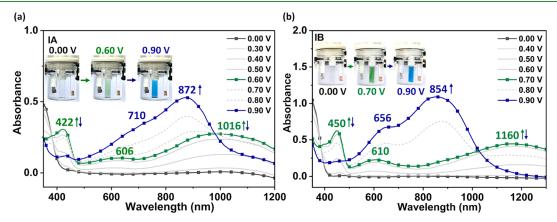


Figure 5. Absorption profiles and electrochromic behaviors of PAs (a) IA (film thickness: 250 ± 30 nm) and (b) IB (film thickness: 250 ± 30 nm) measured on ITO-coating (0.6 cm \times 2.5 cm) glass substrate in 0.1 M TBAP/MeCN.

Table 2. EC Switching Stability of PA IA at 0.6 V in Different Cycles

cycles ^a	$\Delta \mathrm{OD}_{1016}^{b}$	$\Delta T (\%)^c$	$Q (mC/cm^2)^d$	$\eta (cm^2/C)^e$	decay (%) ^f
1	0.26	45.5	1.60	162.5	0.0
2000	0.26	45.1	1.60	162.5	0.0
4000	0.25	43.7	1.54	162.3	0.0
6000	0.25	43.7	1.54	162.3	0.0
8000	0.25	43.2	1.54	162.3	0.0
10,000	0.24	42.5	1.49	161.1	1.0
12,000	0.23	41.8	1.45	158.6	2.0
14,000	0.23	41.3	1.45	158.6	2.0
16,000	0.22	40.3	1.41	156.0	4.0

^aTimes of cyclic scan by applying potential step: 0.0 V 21C6 0.6 V (V vs Ag/AgCl). ^bOptical density change at 1016 nm. ^cOptical transmittance change at 1016 nm. ^dEjected charge is determined from in situ experiments. ^eColoration efficiency is derived from the equation $\eta = \Delta OD/Q$. ^fDecay of coloration efficiency after cyclic scans.

anisole group (IA-MU) resulted in a significant increase in the HOMO energy to -4.52 eV, attributed to the electron-donating effect. Conversely, adding methoxy substituents to the π -bridge (IB-MU) caused the HOMO energy to decrease to -4.64 eV, which aligns with the trends observed in the aforementioned electrochemical results. To clarify this phenomenon, we observed the corresponding dihedral angles of the model units, and these results indicated that the higher oxidation potential of IB-MU was attributed to the orthosubstituents' steric hindrance, which weakens the electron-donating ability of each nitrogen center.

Spectroelectrochemical Behaviors. The spectroelectrochemical spectra of IA are presented in Figure 5a. The resulting film appeared transparent and colorless in the neutral form (0 V). When the applied voltage increased from 0 to 0.6 V, a new absorption peak emerged at 422 nm, along with a broad absorption band around 1016 nm in the near-infrared (NIR) region, and the film color changed from colorless to grass green. These spectral changes in the visible range were attributed to the formation of a stable monocation radical at the TPA center of the N,N,N',N'-tetraphenyl-p-phenylenediamine (TPPA) moiety. The broad NIR absorption band corresponds to intervalence charge transfer (IV-CT) excitation, involving electron transfer from the active neutral nitrogen atom to the cation radical nitrogen center in the TPPA moiety.²⁵ As the applied potential further increased to 0.9 V, the absorption bands corresponding to the monocation radical began to diminish, and a new broad absorption band appeared at 872 nm, accompanied by a color change from grass green to sky blue (Figure S18a). The disappearance of the previous NIR band is attributed to the further oxidation of the monocation radical species, resulting in the formation of dications in the TPPA segments. The UV-vis-NIR absorption changes observed in the IA film at various applied potentials were fully reversible and were accompanied by distinct color transitions. The spectroelectrochemical spectra of IB are presented in Figure 5b, which displayed spectroelectrochemical spectra and color changes similar to those of (Figure S18b) as IA because of the isomeric electroactive chemical structures. However, IB revealed a significantly longer λ_{max} of nearly 150 nm shift and a broader NIR absorption at the first oxidation stage compared to IA. Red-shifted absorption wavelength in the NIR region is ascribed to a weaker electronic coupling between two redoxactive centers that could be attributed to the steric hindrance of the ortho-substituents (methoxy groups) on the π bridge,

hindering the resonance ability of the electroactive centers for ${\bf IB}^+$.

IA and IB exhibit notable optical contrast in the visible and NIR regions. The transmittance change (ΔT (%)) for IA reached 44.7% at 1016 nm and 70.2% at 872 nm in the first and second oxidation states, respectively. Similarly, IB showed a ΔT of 68.3% at 1160 nm and 91.5% at 854 nm during the first and second oxidation stages, respectively. These exceptional absorbance properties in both the visible and NIR regions highlighted the significant potential of IA and IB for dual-band electrochromic applications.

Electrochromic Switching Behaviors. The EC switching stability and response times of the polyamides were assessed by monitoring the absorption changes during potential step applications in kinetic studies. The switching data for the PAs IA and IB are presented in Figure S19. The electrochromic CE ($\eta = \Delta \text{OD/Q}$) and injected charge after different controlled switching cycles were monitored and are summarized in Tables 2, 3, S4, and S5. The results for ΔT ,

Table 3. EC Switching Stability of PA IB at 0.6 V in Different Cycles

cycles ^a	$\Delta \text{OD}_{1160}^{b}$	$\Delta T \ (\%)^c$	$Q (mC/cm^2)^d$	$\eta (cm^2/C)^e$	decay (%)f
1	0.38	58.4	0.68	560	0.0
1000	0.33	53.3	0.65	509	9.1
2000	0.32	52.1	0.65	492	12.1
3000	0.31	51.5	0.64	490	12.5
4000	0.30	49.9	0.64	469	16.3
5000	0.29	49.1	0.63	465	17.0
6000	0.27	46.4	0.60	452	19.3
7000	0.27	46.3	0.60	450	19.6
8000	0.26	45.2	0.59	442	21.1

^aTimes of cyclic scan by applying potential step: $0.0 \text{ V} \leftrightarrows 0.6 \text{ V}$ (V vs Ag/AgCl). ^bOptical density change at 1160 nm. ^cOptical transmittance change at 1160 nm. ^dEjected charge is determined from in situ experiments. ^eColoration efficiency is derived from the equation $\eta = \Delta \text{OD/Q}$. ^fDecay of coloration efficiency after cyclic scans.

CE, and the percentage of CE decay are illustrated in Figure 6. The gradual decline in electrochromic performance during repeated switching cycles could be attributed to several factors. First, repeated oxidation and reduction of the resulting polymers may lead to irreversible side reactions of radical species, such as degradation or coupling, resulting in fewer active redox centers. Second, repeated structural reorganization of the polymer backbone could cause conformational fatigue, which impairs charge delocalization and transfer efficiency. In

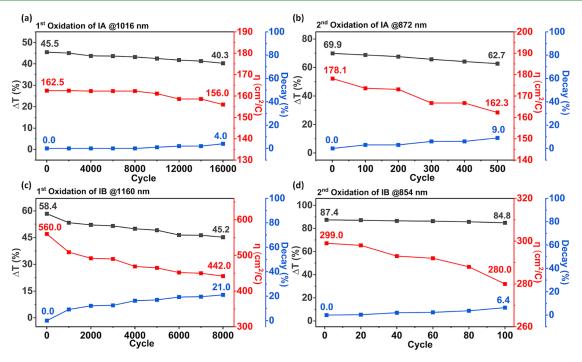


Figure 6. Potential step absorptiometry and current consumption of PAs. Plots of ΔT , CE, and the decay percentage of **IA** for (a) the first oxidation state at 1016 and (b) the second oxidation state at 872. (c) Plots of ΔT , CE, and the decay percentage of **IB** for the first oxidation state at 1160 and (d) the second oxidation state at 854, and all the results were measured in MeCN with 0.1 M TBAP as the supporting electrolyte.

addition, electrolyte infiltration or interfacial instability may weaken the charge injection and redox activity. These combined effects decrease ΔT , lead to longer response times, and reduce the CE.

During the first EC switching test (between 0.0 and 0.6 V), IA showed outstanding switching stability with gratifying reversibility of only 4.0% decay of the CE (156.0–162.5 cm²/ C) at 1016 nm over continuous 16,000 cycles (Figure 6a). This exceptional stability could be attributed to the stable electrochemical properties upon the redox process of IA and also the strong adhesion between the polymer film and the ITO substrate. For the further oxidation switching between 0.0 and 0.9 V, IA revealed relatively poorer switching stability and reversibility than those in the first oxidation state, with 9.0% decay of the CE (162.3-178.1 cm²/C) at 872 nm over continuous 500 cycles (Figure 6b). For IB, switching stability and reversibility of the first oxidation stage between 0 and 0.6 V exhibited 21.1% decay of the CE $(560.0-442.0 \text{ cm}^2/\text{C})$ at 1160 nm over continuous 8000 cycles (Figure 6c). IB exhibited relatively inferior switching stability and reversibility for further oxidation switching between 0.0 and 0.9 V, with a 6.4% decay of CE $(280.0-299.0 \text{ cm}^2/\text{C})$ at 854 nm over continuous 100 cycles (Figure 6d).

IA demonstrated excellent stability at 1016 and 872 nm in the first and second EC processes, respectively, as summarized in Tables 2 and S4. Furthermore, the response times for IA, calculated at 90% of the full ΔT , demonstrated 3.9/1.1 s (coloring/bleaching) at 1016 nm and 5.0/0.9 s at 872 nm for the first and second EC stages, respectively. IB demonstrated a 4.3/0.9 s response time at 1160 nm and 3.4/2.5 s at 854 nm for the first and second EC stages. In summary, PAs IA and IB switched rapidly at the first and second stages.

Steric and Resonance Effects on Electrochromic Durability. The EC stability of the redox-active polyamides for long-term switching between the oxidized and neutral

states is essential for practical application. Two key factors influence the radical stability: first, the electron-donating or -withdrawing ability of substituents; second, the resonance effect of electron delocalization across the electroactive nitrogen centers, both of which significantly impact EC stability. The electrochemical stability between structure-related polyamides of IA, IB, and the corresponding Ib was compared and is summarized in Table 4.

Table 4. Long-Term Stability of the Prepared PAs

polyamide	potential (V)	$\eta (cm^2/C)^a$	cycles ^b	$\Delta T (\%)^c$	decay (%) ^d
IA	$0.0 \leftrightarrow 0.6$	163	16,000	45.5	4.0
IB	$0.0 \leftrightarrow 0.6$	560	8000	58.4	21.1
Ib ^e	$0.0 \leftrightarrow 0.7$	388	10,000	54.0	4.89

^aInitial coloration efficiency. ^bTimes of the cyclic scan by applying a potential step. ^cInitial contrast of the optical transmittance change at 1016 nm for IA, 1160 nm for IB, and 433 nm for Ib. ^dDecay of the coloration efficiency after cyclic scans. ^eThe data were cited from the Ib in ref 15.

The trend of the electrochemical stability in the first oxidation stage is $IA^+ > Ib^+ > IB^+$. IA exhibits an EC stability significantly higher than that of Ib in the first oxidation stage. Since cation radicals are considered to be electron-deficient, stabilizing cation radicals from the electron-donating methoxy group could be expected. Consequently, the higher EC stability of IA could be ascribed to the more electron-donating methoxy substituents in the repeating unit than Ib, resulting in superior EC stability. Intriguingly, Ib exhibits EC stability significantly higher than that of IB in the first oxidation stage, even though IB contains more electron-donating methoxy substituents in the repeating unit than Ib and with the same number of electroactive nitrogen centers and could be expected to form a more stable IB^+ . However, IB^+ has lower stability than Ib^+ ,

which might be caused by the relatively high steric hindrance of the ortho-substituents (methoxy groups) at the electroactive nitrogen centers, hindering the electron delocalization between the two electroactive centers. These results suggest that the resonance ability of the electroactive nitrogen center is far more imperative than that of the electron-donating group to stabilize cation radicals. Furthermore, although IA and IB contain the same number of electron-donating methoxy substituents, IA⁺ displayed significantly higher stability than IB⁺, also attributed to the relatively high steric hindrance of the ortho-substituents (methoxy groups). Notably, the resonance effect of the electroactive centers plays a key role in stabilizing the cation radicals.

The long-term switching stability of polymers is crucial for practical applications. Figure 7 and Table S6 illustrate the

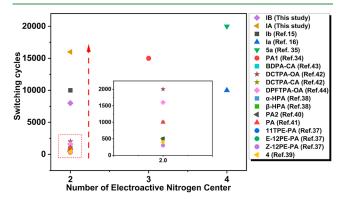


Figure 7. Percentage decay of coloration efficiency (CE) of triarylamine-based polymers after repeated switching cycles at the first oxidation potential.

relationship between the number of electroactive nitrogen centers and the electrochromic stability in the first oxidation stage. In general, polymers with a higher number of electroactive nitrogen centers exhibit better electrochromic stability. Notably, polymer IA investigated in this study, despite containing only two electroactive centers, demonstrated exceptional electrochromic stability, with only a 2% decrease in coloration efficiency after 12,000 switching cycles. This result highlights the excellent stability of IA and emphasizes its synthetic advantage by avoiding the structural complexity often associated with the incorporation of multiple electroactive nitrogen centers.

CONCLUSION

Two novel electrochromic polyamides, IA and IB, bearing dual electroactive nitrogen centers and methoxy-substituted triphenylamine (TPA) units, were successfully synthesized from 4,4'dicarboxydiphenyl ether and newly designed diamine monomers. This isomeric structural difference results in distinct electrochromic (EC) properties. Polyamide IA exhibited superior EC performance, including a rapid switching time (3.9/1.1 s at 1016 nm) and exceptional long-term stability with only 4.0% CE decay after 16,000 cycles. However, it has a smaller transmittance contrast ($\Delta T = 45.5\%$ at 1016 nm) and coloration efficiency value. In contrast, IB with the methoxysubstituted groups on the π bridge reveals higher absorptivity in the visible and NIR region, higher transmittance contrast ($\Delta T = 58.4\%$ at 1160 nm), and remarkable coloration efficiency up to 560 cm²/C. This finding indicates that the methoxy substitution on the π bridge could demonstrate

stronger auxochrome ability. Nevertheless, **IB** showed slower switching, a broader and red-shifted NIR absorption, and a higher CE decay of 21.1% after 8000 cycles, likely due to steric hindrance and a large dihedral angle (\sim 52°) that disrupted π -conjugation and hindered effective resonance stabilization of the cation radicals.

Notably, when compared to a previously reported reference polyamide with similar TPA-based backbones, IA demonstrated comparable or even superior EC stability, despite having a simpler structure with only two electroactive nitrogen centers. Although structurally more complex, the reference polymer exhibited a similar first-stage absorption profile and dual-stage redox behavior but required more extensive conjugation or additional electroactive units to achieve long-term stability. These results underscore the effectiveness of substituent positional control as a rational molecular design strategy to enhance EC performance without increasing synthetic complexity. Overall, this study reveals a new structure—property relationship that advances the development of durable and high-efficiency EC polymers.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsapm.5c01973.

Experimental section, FT-IR spectra, ¹H and ¹³C NMR, TMA, DMA, inherent viscosity and solubility properties, electrochemical properties, and single-crystal information (PDF)

Crystallographic data for 3 (CCDC 2448143) (CIF)

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Author Contributions

Yaw-Terng Chern conceived the idea, designed the experiments, and wrote the manuscript with help and feedback from all authors. Chien-Cheng Yen: data curation and investigation. Yong-Hsien Lin: data curation and investigation. Guey-Sheng Liou: methodology, designed the experiments, and revised the

manuscript. Yu-Jen Shao revised and organized the manuscript Yu-Ting Kao conducted the experiments and analyzed the results.

Notes

The authors declare no competing financial interest.

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